# A.Delta Air Lines

## SUBMISSION TO THE NATIONAL TRANSPORTATION SAFETY BOARD

UNCONTAINED ENGINE FAILURE DELTA **AIR** LINES, FLIGHT 1288 PENSACOLA, FLORIDA JULY 6, 1996 **MC** DONNELL DOUGLAS MD-88, N927DA

This submission contains Delta's analyses, findings, and recommendations derived from information developed during the investigation of this accident. This report is intended to **be** used **as** the basis for final recommendations to prevent this type of accident from occurring *again.* **Areas** addressed include: **(1)** the effects of abusive machining on titanium **alloy** material, **(2)** continued airworthiness certification criteria, **(3)**  standardization and adequacy of guidance from **origlnal** equipment manufacturers for the cleaning and processing of parts prior to visual inspection, and **(4)** supplemental inspection methods to ensure continued airworthiness.

### **CONTENTS**

 $\sim$   $\sim$ 



 $\mathcal{A}^{\mathcal{A}}$  and the proposition of the  $\mathcal{A}$ 

**Barriotti Tit** 

 $\bar{z}$ 



. The construction corresponding to  $\sim$ 

**IrnR** m

<span id="page-3-0"></span>

APPENDIX **A** 

APPENDIX B

APPENDIX C

APPENDIX D

**mllmn** r., .\_ ..-. ....

#### **EXECUTIVE SUMMARY**

<span id="page-4-0"></span>On July 6, 1996, Delta Air Lines flight 1288, a McDonnell Douglas MD-88, experienced an uncontained failure of the left engine during the beginning of the takeoff roll on runway 17 at the Pensacola Regional Airport, Pensacola, Florida. Subsequent investigation revealed that the first stage fan hub of the left JT8D-219 engine had fractured, resulting in an uncontained failure of the engine. Fragments from the engine penetrated the aft fuselage, causing fatal injuries to two passengers and substantial damage to the aircraft.

Investigation of this accident has revealed that the accident fan hub, which had accumulated approximately 14,000 cycles, fractured from an existing low cycle fatigue crack that had originated near an area of smearing, or "scuff marks", on the inner surface of a tie bolt hole. The surface anomalies were determined to be a microstructural deformation of the titanium alloy resulting from abusive machining (drilling) during the manufacturing process. **A**  metallurgical examination of the fracture surfaces indicated that the crack had originated shortly after the fan hub was put in service. The certified safe life, or mandatory part replacement time, for the -2 19 fan hub is 20,000 cycles.

Delta has determined that the probable cause of this accident was the catastrophic failure of the left engine first stage fan hub as a result of a low cycle fatigue crack that originated from a microstructure defect created by abusive machining during the manufacturing process.

The safety issues in this report include the effects and detection of abusive machining on titanium alloy material, continued airworthiness certification criteria, standardization and adequacy of guidance from original equipment manufacturers for the cleaning and processing of parts prior to visual inspection, and supplemental inspection methods to ensure continued airworthiness .

Safety recommendations are included in this report, and are intended to be used as the basis for final recommendations to prevent this type of accident from occurring again.

#### <span id="page-5-0"></span>**1. FACTUAL INFORMATION**

#### **1.1 Flight History**

On July 6, 1996, Delta Air Lines flight 1288, a McDonnell Douglas MD-88, experienced an uncontained failure of the left engine during the beginning of the takeoff roll on runway 17 at the Pensacola Regional Airport. The engines were just reaching peak thrust when the flight crew heard a noise and rejected the takeoff. The aircraft was brought to a stop on centerline approximately 13 50 feet from the runway threshold.

Subsequent investigation revealed that the first stage fan hub of the left engine had fractured, resulting in an uncontained failure of the engine. Fragments from the engine had penetrated the aft fuselage, causing fatal injuries to passengers and substantial damage to the aircraft. One effect of the damage was a total electrical failure that precluded crew communication via either the interphone or public address system.

There were 5 crew members and 137 passengers on board at the time of the event. Passengers that occupied seats in the aft portion of the cabin were evacuated immediately from the aircraft; those forward in the cabin remained seated and were later deplaned via airstairs.

Emergency medical service (EMS) and airport rescue fire fighting (ARFF) personnel responded to the aircraft after being alerted by the airport tower.

#### **1.2 Injuries**

Two passengers, occupying seats in a row adjacent to the failed engine, were fatally injured Another passenger, seated in the same row, was seriously injured. Two other passengers sustained minor injuries during the evacuation.

#### **1.3 Damage**

The uncontained failure of the left engine was caused by a fracture of the first stage fan hub, that in turn caused various components and other pieces to separate from the engine. Many fragments, including fan blades and a section of the fan hub, penetrated the aft fuselage and entered into the cabin. Other engine components, including the nose cowl, were scattered on the runway behind the aircraft starting from the point of failure about 385 feet from the threshold. Debris, including two large sections of the fan hub, was found in areas off both sides of the runway, at various distances from the centerline.

The aircraft sustained extensive skin and structural damage to the aft left side of the fuselage immediately adjacent to the engine fan section. Fragments penetrated the hselage and passed through the aft cabin area, causing damage to interior furnishings. as well as to fuselage skin and structure on the right side.

<span id="page-6-0"></span>*An* electrical power feeder bundle from the number 2 engine generator that runs above the cabin ceiling along the upper right interior of the fhselage was almost entirely severed.

The left engine was destroyed.

#### **1.4 Engine Information**

The **MD-88** is equipped with the Pratt & Whitney JT8D-219 engine, which is an axial-flow front turbofan engine having a fourteen stage split compressor section. The engine fan, the first stage of the compressor section, consists of **34** blades attached to a titanium alloy fan hub.

The left engine had a total operating time of 7371.7 hours and 5905 operating cycles at the time of the accident. It had been installed on the accident aircraft on January 1, 1996, and had subsequently accrued 1528 hours and 1142 cycles.

#### **1.5 Fan Hub Information**

The fan hub weighs approximately 150 pounds, and is forged from the titanium alloy 6AL4V, which contains 6% aluminum and 4% vanadium. The major portion of the hub consists of a disk forging that holds the fan blades in dovetail slots around the outer radius. Integral to the disk is a cone shaped nose piece that engages the front end of the hub to the inner race of the number 1 engine bearing. The aft end of the hub attaches to the following compressor component with 24 tie bolts that pass through holes in the rim of the disk just inside of the dovetail slots. Evenly spaced between each tie bolt hole are 24 stress redistribution holes. Each hole through the disk is approximately 3 inches in depth.

The accident fan hub had a total of 16,542 hours and 13,835 cycles at the time of the accident, and 1142 cycles since its last fluorescent penetrant inspection (FPI) and visual inspection by Delta Air Lines. Records indicate that the fan blades had been replaced prior to the engine being installed on the accident aircraft on January 1, 1996.

#### **1.5.1 Metallurgical Investigation**

Post-accident investigation revealed that the accident hub fractured in two places radially on the rim circumference as well as longitudinally in the cone area. One of the rim radial fractures emanated from a pre-existing fatigue crack in one of the tie bolt holes. According to the NTSB Metallurgist's Factual Report, the fracture contained fatigue striations that originated from points near the aft end of a tie bolt hole interior surface. Fatigue cracking had propagated approximately 1.5 inches radially inboard. Beyond that, fracture features were typical of an overstress separation.

**A** striation count on the fracture face using a scanning electron microscope (SEM) revealed the total number of fatigue striations was roughly equivalent to the total number of flight cycles for the fan hub. The number of striations, along with the appearance of the fracture

**Page** 2

and company

<span id="page-7-0"></span>surface, suggested that the crack was present on the aft face of the hub (0.46 inches) and along the wall of the tie bolt hole (0.9 inches) at the time the last fluorescent penetrant inspection (FPI) accomplished by Delta in December 1995.

The pre-existing fatigue crack originated near smearing, or "scuff marks", in the tie bolt hole. Small, shallow surface chips were noted in the smeared areas. The surface of the smeared areas, exhibited evidence of circumferential machining marks, probably from the boring operation during original part manufacture. The remainder of the hole surface away from the scuff marks exhibited a cross hatched pattern, typical of a honing operation.

Metallographic examination indicated that the microstructure of the metal along the surface of the hole wall adjacent to the fracture location was severely deformed and contained numerous secondary cracks.

#### **1.6 Hub Manufacturing Information**

Volvo Flygmotor in Trollhatten, Sweden, is the manufacturer of fan hubs for Pratt & Whitney JT8D-200 series engines. The accident hub was manufactured by Volvo in 1989; Pratt & Whitney subsequently received the hub and installed it on an engine that was delivered to Delta on April 27, 1990.

The -219 titanium fan hubs are forged in the United States and then delivered to Volvo where they are machined, finished, and inspected. During manufacturing, the tie bolt and stress redistribution holes in the disk rim are created using a three-phase process consisting of drilling, boring, and honing. After manufacturing is complete, the hubs undergo various dimensional and non-destructive tests to ensure Pratt & Whitney specifications are met.

#### **1.6.1 Coolant Channel Drill**

Manufacturing records for the accident hub indicate that a coolant channel drill in a computer controlled machining center was used to drill both the tie bolt and stress redistribution holes in the hub rim. A coolant channel drill permits coolant to flow into the hole just behind the carbide cutting edges during the drilling process. The coolant entering the hole serves as a drilling lubricant to reduce heat and as a flushing agent for removing drill chips from the hole.

According to manufacturing records<sup>1</sup>, the request to change the manufacturing process to use a coolant channel drill was submitted by Volvo and subsequently granted engineering approval by Pratt & Whitney in February 1988. The Pratt & Whitney Process Approval Record indicated that the replacement of the drill type was considered an "insignificant" change.

According to Volvo, the change to the coolant channel drill was implemented because it permitted a straighter hole to be drilled, thus better maintaining drilling tolerances. However,

NTSB Public Hearing, Exhibit *SG* 

<span id="page-8-0"></span>the use of the coolant channel drill was discontinued shortly after 1989 reportedly due to hole oversizing.

#### **1.6.2 Blue Etch Anodize Process**

After all machine work is completed on a hub, the part undergoes an inspection phase that includes the blue etch anodize (BEA) process. The BEA inspection is unique to titanium, and consists of a visual inspection of an anodized surface for unique patterns that indicate anomalies associated with certain microstructure changes in the parent metal.

Volvo manufacturing records of the accident hub indicate that when the BEA inspection procedure was accomplished, mechanical marks were detected inside the tie bolt hole where the fatigue crack originated. The BEA procedure in effect at the time the accident hub was manufactured did not have any failure criteria that would have identified the microstructural changes present in the tie bolt hole. Based on the BEA inspection results, the hub was referred to a visual inspection process, where, because the part satisfied all Pratt & Whitney inspection criteria, the marks were accepted.

Post-accident investigation identified seven fan hubs that were found to have various unusual conditions detected inside the holes during the Volvo BEA inspection process. These hubs, including the accident hub, entered service after being certified as meeting BEA and visual inspection criteria in effect at the time.

#### **1.7 Hub Certification**

The accident fan hub was certified by the FAA as meeting the requirements of FAR 33.14, Start-Stop Cyclic Stress (Low Cycle Fatigue) and FAR 33.4, Instructions for Continued Airworthiness, and supplemented by Appendix A33.4, Airworthiness Limitations Section. $^3$ 

#### **1.7.1 Safe Life Certification**

The safe life limit (mandatory part replacement time) for the JT8D-200 series fan hub is 20,000 flight cycles in compliance with FAR A33.4. This information is found in the Pratt & Whitney JT8D Engine Manual, Chapter 05-10-00, entitled "Time Limits at Normal Take-off Thrust Rating".

#### **1.8 Operator Non-destructive Inspections**

According to Delta's maintenance records, the fan hub underwent **a** fluorescent penetrant inspection (FPI) at the base maintenance facility in Atlanta in October 1995 after removal from the engine for blade and restricted parts removal. FPI is a process involving the submersion of a part into a low viscosity fluorescent dye bath, followed by a pre-rinse, emulsified by washing with a high viscosity solution, and final rinsed. After the application of

 $2$  NTSB Public Hearing transcript

<sup>&</sup>lt;sup>3</sup> NTSB Public Hearing, Exhibit 8K

<span id="page-9-0"></span>a developer, the fluorescent dye, which is retained by cracks and other surface defects through capillary action, luminesces under ultra violet *(UV)* light inspection. At the time of the FPI, the hub had accrued 12,693 flight cycles. Subsequent to the inspection, the hub was installed on the accident engine on December 29, 1995.

Delta's procedures for the cleaning, preparation and non-destructive inspection of parts are governed by standard practices that are based on procedural recommendations provided by the original equipment manufacturers (OEMs). For the titanium -219 hub involved in the accident, the Delta process standards and Pratt & Whitney service process operating procedures (SPOPs) pertaining to the cleaning and FPI processes are as follows:



Delta's procedures pertaining to similar parts manufactured by different OEMs may vary because of differences in the OEM recommended procedures. That is, the procedures that cover the cleaning, processing and inspection of a given part manufactured by Pratt & Whitney may differ from the procedures that cover a similar part manufactured by General Electric or Rolls Royce. Delta's process standard may deviate from a particular manufacturer's recommendation when there is inconsistency or disagreement between the recommended procedures of the various OEMs on how to process a similar part. For example, consistent with the recommended procedures published by General Electric and Rolls Royce, Delta's process standard does not require a titanium part to be spray rinsed following a plastic media blast, even though a spray rinse is recommended, if necessary, by Pratt & Whitney. On occasions where Delta's process standard deviates significantly from the recommended procedures of the OEM, Delta will coordinate those differences with the OEM to ensure that the integrity of the process is not compromised. **A** summary comparison of Delta's process standards for cleaning, media blasting, and FPI with the recommended procedures from three OEM's is provided in Appendices **C** and D.

#### **1.8.1 Cleaning**

In preparation for the FPI, the hub was cleaned per Delta Process Standard (P.S.) 900-1-1, No. 18. The process standard covers the approved materials and procedures for cleaning dry film lubricant, and carbon removal and paint stripping of various aircraft and engine parts including aircraft wheels, landing gear parts, seat parts, and miscellaneous engine parts manufactured by Pratt & Whitney. The process standard is based on the original equipment manufacturer's (Pratt & Whitney) recommended procedures for cleaning titanium parts and specifies the sequence in which the parts are immersed in various alkaline solutions, rinsed, and dried prior to a plastic media blast and subsequent inspection. The specific alkaline solutions used and their concentrations, dwell times in the solutions, and rinse times used for <span id="page-10-0"></span>the cleaning and drying of titanium parts are based on the manufacturer's recommended procedures.

The Delta cleaning process for titanium hubs manufactured by Pratt & Whitney in effect in October 1995 included three stages. First, the hub was placed in a wire mesh basket and dipped in light duty degreaser for a specified dwell time. It was then spray-rinsed with city tap water, followed by a cold water dip rinse. In the second phase, the hub was dipped in a more aggressive alkaline cleaner for a specified dwell time, after which it was spray-rinsed and dipped in a cold water rinse. Finally, the hub was dipped in a third alkaline solution which softens dry lubricant and anti-gallant compounds. Following this last alkaline solution dip, the hub was dipped in a hot water bath for "flash drying."

Flash drying is an evaporative drying technique that involves dipping a part in a hot water rinse until the part is heated to the same temperature as the surrounding water. When the part is lifted out of the water, the water on the surface quickly evaporates as a result of the high temperature. Delta's process standard 900-1-1 No. **18** states that the part should be immersed in hot water (150-200 $^{\circ}$ F) until the temperature of the part equals the temperature of the water. Pratt & Whitney's Standard Practice SPOP 209, upon which Delta's process standard is based, states that the part should be immersed in hot water (I 50-200°F) until the temperature of the part is at the water temperature to flash dry. The cleaner relies on a visual inspection of the evaporation of water on the surface of the part to determine that a part is adequately dry. There is no published guidance or tools available to the cleaner to determine whether water that may be present in cracks or other surface anomalies has fully evaporated using the flash drying technique.

#### **1.8.2 Media Blast**

After the hub has been cleaned and flash dried, it is subjected to plastic media blast to remove anti-gallant per Delta P.S. 900-1-1, No. 21. Delta's P.S. 900-1-1, No. 21 covers the materials and procedures for cleaning aircraft and engine parts using dry plastic media (Exhibit 11S). The purpose of a media blast is to remove residual surface contaminants such as heat scale, carbon deposits, corrosion and rust. Delta's process standard for media blasting the hub involved in the accident is based on Pratt & Whitney Specification SPOP  $19<sup>4</sup>$ , which describes part preparation, air pressure and blasting technique to ensure the part is cleaned. Following the media blast, SPOP 19 and P.S. 900-1-1, No. 21 both state that the part should be blown clean with air. Delta's procedure then calls for an inspection of all surfaces and cavities to ensure no blasting media is entrapped or remaining on the part, followed by dipping the part in a rust preventive solution. Pratt & Whitney's procedure recommends the part be cleaned through a pressure spray rinse, if necessary, after the part is blown dry and before the corrosion inhibitor is applied. **As** noted earlier, Delta's procedure does not call for a water rinse following the plastic media blast.

NTSB Public Hearing, **Exhibit** 11L

#### <span id="page-11-0"></span>**1.8.3 Fluorescent Penetrant Inspection**

Once the titanium hub has been cleaned, it subjected to a multi-step process to prepare it for fluorescent penetrant inspection. The part is first dipped in a vat containing fluorescent penetrant dye for a minimum dwell time of 30 minutes. Mer the hub is removed from the dye vat, it is spray-rinsed with city tap water for a period of one to two minutes. The hub is then dipped in a vat containing emulsifier for a period of 30-90 seconds. The purpose of the emulsifier is to break down fluorescent dye that is present on the surface of the part, thereby enhancing the visual contrast of fluorescent dye that penetrates any surface defects. The dwell times in the dye penetrant and emulsifier solutions are specified in Delta's Process Standard P.S. 900-6-3 No. 02, and are based on recommendations from the original equipment manufacturer. Upon removal from the emulsifier, the part is spray-rinsed with city tap water for a period not longer than one to two minutes in any one area. The purpose of the emulsifier is to render the surface fluorescent water washable thereby enhancing the visual contrast of fluorescent dye that penetrates any surface defects. The hub is then placed in a drying oven at 140-160 degrees Fahrenheit to facilitate surface drying. After removal from the drying oven, the processor visually inspects the part to ensure that it is dry, Once it is determined that the part is dry, a dry powder developer is applied a low pressure through a hose nozzle. The developer is sprayed on the exterior and interior visible surfaces by sweeping the hose nozzle in various directions over the hub. After application of the developer, the hub is ready for inspection.

The fluorescent penetrant inspection takes place in a darkroom, called a "tent," that is enclosed by walls and a ceiling made of a heavy canvas material. The tent contains an overhead white light, overhead UV light, and hand-held incandescent and W lights. The overhead white light and hand-held incandescent light are used to perform visual inspections prior to the UV inspection and in the event of a positive indication during the UV inspection. The UV lights are used to illuminate any fluorescent dye that is remaining on the surface of the part or that has penetrated surface defects through capillary action. **A** crack or similar defect on the surface of the part will appear under UV light as a "glowing" greenish-yellow indication. Magnifying lenses and mirrors are available to the inspectors for closer examination of potential positive indications picked up during inspection under UV or incandescent lighting.

#### **1.8.4 Post-Accident Observations**

During a visit to Delta's FPI line following the accident, the inspection of a titanium hub similar to the one involved in the accident was observed by the maintenance group representatives on two separate occasions. The first observation was accomplished with a regular shop FPI inspector performing the task. The second was done by the individual who gave the accident hub its last FPI. The FPIs observed by the group members were essentially the same except for the individual inspector techniques. The parts are placed on a set of plastic rollers that allow it to be moved relatively easy once in the tent. **A** round shaped work area having the same type rollers is the inspector's work area. The tent is provided with an

<span id="page-12-0"></span>overhead hoist and strap for manipulating the part during inspection. This tooling aid is used at the inspectors' discretion.

During the inspections, the group observed that the UV spotlight cannot effectively penetrate into the tie bolt holes when viewed from the top of the hub. The geometric design of the -2 19 hub, especially the conical front, and the size of the floodlight prevents the positioning of the spotlight in such a way that both (1) the light penetrates deeply into the tie bolt holes, and *(2)*  the interior of the hole can be seen by the inspector if any developer was in the hole. On the bottom of the hub, proper alignment of the spot light in the bore holes can only be done if the inspector rotates the hub under a very specific frame of reference. Instead, the inspector was observed to rotate the hub in approximate 120 degree increments and then physically move his body and light along an arc to inspect the holes. During the visual inspection, the first inspector said that finding cracks in the bolt and balance weight holes is not really possible. The second inspector also stated to investigators that it was very difficult to see into the holes on the hub.

#### **1.9 NTSB Recommendations**

As a result of the initial investigation of this accident, in July 1996 the NTSB issued four recommendations for FAA action. In summary, the NTSB recommended that the FAA:

- Require (within 500 cycles) an eddy current inspection of JTSD-200 fan hub tie bolt and  $\bullet$ stress redistribution holes to be performed on those hubs that have accumulated more than 10,000 cycles.
- $\bullet$ Require an FPI and eddy current inspection of JTSD-200 fan hub tie bolt and stress redistribution holes to be performed on all hubs by a fixed number of cycles, as determined by the risk of crack propagation from manufacturing flaws.
- Review and modify the processes by which Volvo and Pratt & Whitney permitted JT8D- $\bullet$ 200 series fan hubs to be placed in airline service following indication of mechanical damage in the tie bolt holes based on a BEA inspection.
- Review and revise, in conjunction with engine manufacturers and air carriers, procedures,  $\bullet$ training, and supervision of inspectors for performing FPI and other non-destructive testing of high energy rotating engine parts, with emphasis on the JTSD-200 series tie bolt and stress redistribution holes.

#### **1.10 FAA Airworthiness Directives**

and a real

The FAA issued a Priority Letter Airworthiness Directive (AD) in July 1996 that identified, and directed immediate removal from service, the remaining six JT8D-200 series fan hubs that had recorded indications of tie bolt hole surface anomalies found during the manufacturer's **BEA** inspection process.

In addition, another AD was issued in February 1997 that prescribed a continuing inspection program for tie bolt and stress redistribution holes in all JT8D-200 series fan hubs based on procedures outlined in a Pratt & Whitney Alert Service Bulletin issued in September 1996.

والمستحدث

<span id="page-13-0"></span>The program mandates that a combination of eddy current and FPI inspections to the hole areas must be accomplished using various inspection schedule options that are predicated on the number of total cycles accumulated on the fan hub.

#### **2. ANALYSIS**

#### **2.1 Microstructure Damage from Machining**

The Metallurgist's Factual Report indicates that the pre-existing fatigue crack originated at "scuff marks" or smearing in one of the fan hub tie bolt holes. Small, shallow surface chips were noted in the smeared areas. The surface of the "scuff marks", or smeared areas, exhibited evidence of circumferential machining marks, probably from the boring operation during original part manufacture. The remainder of the hole surface away from the scuff marks exhibited a cross hatched pattern typical of a honing operation.

**A** scanning electron microscope (SEM) examination revealed that the smeared area at the origin contained numerous parallel cracks (ladder cracks), parallel to the thumbnail area at the fracture origin. Also noted were small shallow chip outs that appeared to be associated with the ladder cracking.

Metallographic examination of a cross section through the fracture origin revealed that the microstructure along the tie bolt hole wall adjacent to the fracture location was severely deformed and contained numerous secondary cracks (previously mentioned ladder cracks). The metallographic examination disclosed that the layer of distorted microstructure adjacent to the fracture face consisted of two zones of depths ,002 inches and *,0035* inches respectively. The microstructure in the damaged zone closest to the hole surface "appeared unclear and heavily layered". This appearance is typical of alpha rich zones in titanium and is similar in appearance to the white surface layer caused either by extensive deformation and/or oxidation as reflected in published literature pertaining to the microstructure of titanium'. The microstructure in the second zone consisted of heavily deformed alpha and beta grains. The microstructure of the base material consisted of equiaxed alpha grains in a transformed beta mix, typical for a titanium base alloy processed below the beta transus temperature.

Abnormal Knoop hardness readings of 581 HK (52 HRC) were measured at a distance of .001 inch in the damaged microstructure area, compared to normal hardness readings of **347** HK **(34** HRC) in the undamaged zone. This change in microhardness was much more severe than published literature indicated it should have been, indicating that previous research quantifying hardness abnormalities associated with "abusive machining" to date is inadequate.

(Abusive machining is defined as any process that deleteriously affects the microstructure of the parent metal.)

**Page** 9

والأسابهما

**MATHEMATICS** 

**Appendix A, 4** *<sup>5</sup>*

The surface damage (smearing) seen in this case, apparently caused by an improper machining operation, is remarkably similar in appearance and hardness to damage noted in the investigation of another fractured fan hub in  $1982<sup>6</sup>$ . In both instances, the surface damage caused by the abusive machining eventually resulted in catastrophic fan hub break up, and uncontained engine failures. The serious nature of the problem is compounded by the fact that, although over a thousand articles have been published on various aspects of titanium and its alloys, there is comparatively little published work on the detailed effects of abusive machining on the microstructure of titanium alloys.

#### **2.1.1 Microstructure Complexity**

Ti-6Al-4V is a high-alpha, lean-beta alloy of titanium; the structure of the alpha phase is hexagonal close packed (HCP) while the beta phase is body centered cubic (BCC). In unalloyed titanium, only alpha exists below  $1620^{\circ}$  F while only beta exists above  $1620^{\circ}$  F. Alloying elements stabilize either of the two phases and so, by selecting suitable elements and proper heat treatment, both phases can be made to exist at room temperature.



The standard composition of Ti-6AI-4V (range or maximums)<sup>7</sup>:

Aluminum stabilizes the alpha phase while vanadium and iron stabilize the beta phase. Oxygen, carbon and nitrogen which are considered impurities, form interstitial solutions and stabilize the alpha phase. With the above composition the beta transus temperature **is** 1825" F  $\pm$  25° F, but changes in the interstitial elements affect the beta transus temperature.

The interstitials also affect the alloy as follows:

01-1" I .I

strength increases with interstitial content at intermediate temperatures up to 500" F without much loss of ductility, and strengthening effect decreases with increasing temperature,

\_-

**Appendix A, 5**  *6* 

<sup>&</sup>lt;sup>7</sup> Appendix A, 1

- <span id="page-15-0"></span>the beta to alpha transformation rate is accelerated by the presence of interstitials dissolved in the beta phase, and
- *0*  the notch sensitivity, especially at low temperatures, is increased with increasing interstitial content.

In general, the microstructures of Ti-alloys are manipulated by heat treatment and manufacturing process to give the desired strength and toughness from the following characteristics: (1) the relative amounts of alpha, beta, and transformed beta (alpha prime), (2) the morphology of the phases (equiaxed or acicular), **(3)** the texture of the grains, and (4) the sub-microstructural constituents (alpha 2 and omega) present in the primary alpha and beta phases. The formation and effects of these characteristics have been studied in detail<sup>8</sup> and are mentioned here to indicate the complexities involved in obtaining a desired composite property.

**A** review of literature' regarding titanium machining shows that relatively few studies have been conducted which examine and quantify the effects of machining, and drilling in particular, on the surface microstructure of titanium alloys (including Ti-6A1-4V).

#### **2.1.2 Machining Titanium**

The potential for microstructural changes during titanium machining can arise from the high cutting temperatures involved and the chemical reactivity of titanium. The high temperatures are mainly due to the tool-titanium high contact pressures, and the poor heat conductivity of titanium. The thermal conductivity of titanium is one third that of steel<sup>10</sup>. Hence the tool-chip interface temperatures when machining titanium are higher than they would be when machining steel. The temperature rise depends on the cutting speed and the tool material: for high-speed-steel tools and cutting speeds of 10-60 fpm, the temperature rise can range from 500" to 850" F, while for carbide tools at speeds from 10-220 fpm, the temperature rise can range from  $400^{\circ}$  to  $1800^{\circ}$  F. These measured temperatures may in fact be lower than the actual, instantaneous, temperature rise at the tool-workpiece interface. The high temperatures therefore are a potential source for microstructural changes in the titanium surface. Hence, any change in the tool material (e.g., from solid HSS to drills with carbide cutting edge inserts), feed rate, or speed of machining (as implemented by Volvo), should be considered as a *significant change*, and therefore require an extensive microstructural characterization<sup>11</sup> prior to implementation.

Compounded with high cutting temperatures is the strong chemical activity of titanium with the tool materials at elevated temperatures. This can macroscopically produce galling, welding, and smearing, since an alloy is continuously formed between the titanium chip and the tool material<sup>12</sup>. This alloy usually passes off with the chip but the potential for smearing on

**Appendix A, 6- 14 8** 

<sup>&</sup>lt;sup>9</sup> Appendix A, 2-4, 15-20

**Appendix A, 15**  *10* 

<sup>&</sup>lt;sup>11</sup> Appendix A, 23

**Appendix A, 15**  12

<span id="page-16-0"></span>the workpiece also exists at very high temperatures. Such smearing has been observed during service experience of drilling the Ti-6Al-4V fan hub, but due to a lack of understanding, has been improperly characterized as machining marks.  $^{13}$ 

Oxidation is another problem at high temperatures due to the high chemical activity of titanium<sup>14</sup>. Typically, oxidation of the surface layers results in a large amount of the alpha phase [Figure 11, whose thickness (depth into the surface) increases with the temperature and the time held at that temperature [Figure **21.** However, since the kinetics of oxidation can also be affected by the applied stress (strain hardening), i.e. an Eyring type dependence on stress and temperature, the rate of oxide formation may be accelerated by high tool-workpiece stress and temperature. This aspect appears not to have been studied, specifically for titanium alloys.

To keep the cutting temperatures down, cutting fluid is used to minimize any tool-chipworkpiece friction and welding. Various types of cutting fluids such as water soluble emulsions, halogenated oils, and synthetic chemical are normally used for machining titanium. However, the use of chlorinated oils should be avoided since chloride residues from these oils may lead to stress corrosion cracking of the parts during high temperature service<sup>15</sup>. Any change in either the coolant type or in its delivery, should require a complete re-evaluation of the machining process including a study of the surface microstructure of the finished part<sup>16</sup>. For example, in the drilling operation of the Ti-6Al-4V fan hub, when changing the drill type from a regular **HSS** drill to a coolant channel drill, the process should not have been considered "approved" until a thorough evaluation of the finished surface had been completed. This was not done on a consistent basis<sup>17</sup>, and is likely the root cause of the accident fan hub failure.

Cross-sectional hardness measurements on Ti-6Al-4V surfaces after machining operations like grinding have shown that if the operation has been gentle, the surface will exhibit a shallow white layer of hardness measured lower (by about 4 HRC) than the bulk material. But when the grinding has been abusive, the surface white layer will be distorted to a depth of 0.005 inches, and will measure an increased hardness (of about 4 HRC) compared to the bulk material<sup>18</sup>. Service experience gained from drilling of the JT-8D fan hub made from Ti-6Al-4V has shown that under abusive conditions, the white layer, or work affected microstructure, can extend to a depth of  $0.025$  inches and have a measured hardness increase of up to 21 HRC<sup>19</sup>. **As** a result, during normal operating stresses, surface cracks can initiate from this brittle layer caused by abusive machining. Such cracking has been observed<sup>20</sup>, but it is not yet known if the cracks were initiated under service load application, or initiated during the abusive machining process.

**Appendix A,** 24 **13** 

**Appendix A, 1,4 14** 

**Appendix A, 15 15** 

**Appendix A, 23 16** 

**Appendix A, 25 17** 

**Appendix A,** 2,3 **18** 

**Appendix A, 5 19** 

**Appendix A, 5**  *20* 

<span id="page-17-0"></span>Extensive surface deformation during abusive machining results in a highly strain hardened microstructure that, because of high machining temperatures, can undergo recovery and recrystallization. As seen from service experience<sup>21</sup>, such recrystallization can result in a large, coarse grained microstructure in a thin surface layer. Because there have been no studies done, there is very little known about the recrystallization temperature and kinetics that lead to this deformed surface layer.

#### **2.1.3 Titanium Drilling**

During titanium drilling<sup>22</sup>, thin chips flowing at high velocities are likely to fold and clog in the flutes of the drill, causing additional frictional temperature rise and smearing. These problems can be minimized by using short sharp drills, supplying cutting fluids to the cutting zone and using low speeds and positive feeds.

The accepted rule when drilling titanium is to keep the drill cutting and never allow the drill to ride in the hole without cutting metal. This is done to prevent temperature build-up, and the attendant oxidation that can occur. It could not be determined if this aspect was considered during the deep hole drilling of the accident hub, but it could account for the altered surface layer that resulted.

**A** sharp drill produces tight-curling chips, but as the drill dulls and the cutting temperature rises, feather-type chips are produced. This is a sign that the drill is dull. The appearance of discolored chips indicates that a drill has failed.

When drilling holes through a part, the general practice is to not drill all the way through on a continuous feed. The drill should instead be retracted and cleaned before breakthrough. The chips should be flushed from the hole before continuing to drill through the part. During coolant channel drilling, chips are flushed out by a coolant that is forced into the hole through a channel in the drill. However, when the drill breaks through at the exit side of the hole, there is no coolant backpressure to continue flushing the chips out. The trapped chips melt and can cause smearing or scuffing in the hole as there is now no coolant available to keep the drillworkpiece interface temperatures down.

**A** minor imperfection on the surface of the part, caused by a mildly abusive machining operation, can prove critical. Imperfections can serve as initiation sites for fatigue and/or stress corrosion, thereby drastically shortening the life of the part. In order to avoid such problems, all changes in manufacturing operations, including machining, should be thoroughly reviewed to determine the microstructural effects on the material, including strength and toughness. **A** review of this type is usually facilitated by the existence of sufficient data showing how changes in the process affect the material. The investigation of this accident reveals that such information is not readily available for certain machining operations on Ti-6AI-4V, including drilling.

**Appendix A, 5 21** 

**Appendix A, 15**  *22* 

#### <span id="page-18-0"></span>**2.1.4 Industry Data**

**A** lack of data exists pertaining to the issue of microstructure abnormalities caused by abusive machining (drilling) during the manufacture of titanium alloy parts, and specifically the susceptibility to microstructure damage from use of the coolant channel drill. Any study and collection of data regarding the effects of drilling on Ti-6AI-4V should include at least the following.

- $\bullet$ Determine temperature rise with and without coolant, and, using the recommended drill types, at different speeds and feeds both in the recommended and abusive conditions. Sampling should include work pieces of varying thicknesses within the extremes of deep hole geometries usually encountered in the manufacture of titanium aircraft and engine components. Additionally, tool-workpiece force/pressures should be measured during the cutting operation.
- Conduct hardness tests, and characterize the morphology, composition, plastic  $\bullet$ deformation and any micro-cracking of the near-surface cross-section by optical microscopy, SEM and TEM and X-ray diffraction methods.
- Study and document post-drilling properties to include tensile strength, number of cycles needed to initiate a fatigue crack of set length as a function of cyclic stress, and time to initiate a stress-corrosion crack of pre-determined length in a NaCl solution as a function of stress.
- Correlate measured properties of strength, fatigue life and stress corrosion susceptibility with surface characterization, and thus with the type of drilling operation.

Currently there is no comprehensive database that contains the effects of drilling changes on the microstructure and properties of Ti-6AL4V; i.e. data that can be used as evaluation criteria for certification and approval when changes in the machining process (drilling) are proposed. The investigation of this accident reveals that changes to the machining and tooling process should always be considered "significant changes", and approval should be substantiated on the basis of equivalency with a rigorously established industry baseline.

#### **2.2 Safe Life Certification**

The 20,000 flight cycle safe life limit for compliance with A33.4 was established by applying a 20% life reduction factor to the analytically established "BO. **1"** (risk factor of 0.1 or less, or for a safe life component, it is estimated that the chance of the strength falling appreciably below the design ultimate value is about 1 in 1000) statistical design life goal of 25,000 flight cycles for crack initiation, with no credit for crack propagation<sup>23</sup>.

Safe life certification is based on the assumption that the part will remain crack free during the service life. It is assumed that the number of cycles to failure (with failure defined as fatigue crack initiation) are related to the "average" strength of the part. The NTSB Metallurgist's factual report<sup>24</sup> indicates that fatigue crack initiation in the accident hub began during the first

**AppendixB,** 1 *<sup>23</sup>*

**<sup>24</sup>**NTSB **Public Hearing, Exhibit 15B** 

<span id="page-19-0"></span>244 cycles of the service life of the part. The possibility also exists that, due to residual stresses associated with the damaged microstructure and a severe hardness gradient<sup>25</sup>, the crack may have initiated prior to the part being put into service. This is obviously contrary to the certification requirement of no fatigue crack initiation during the entire life of the part.

**As** discussed in the analysis above, early fatigue initiation in the accident hub was caused by abusive machining (drilling) during the manufacturing process that resulted in a material defect (deleteriously altered titanium microstructure) going undetected by the supplier's (Volvo) quality assurance program, thus allowing a defective part to enter service. The consequence is that the fatigue endurance limit that was determined by the fan hub certificate holder (by applying a life reduction factor) was grossly exceeded. This indicates the need for additional validation of life reduction factors, or the establishment of a fatigue probabilistic reliability standard for safe life determination.

Such a life reduction factor had been proposed by a commenter during the review of FAR 33.14, and described in Amendment 33-10, Dec. 16, **1983.** The commenter had suggested that a fixed percentage of the predicted life be established as the "initial service life". The suggestion was not incorporated because it was thought that the existing methodology of predicting low cycle fatigue life was adequate, and that adoption of a reduced initial life would "place undue burden on the applicant (engine manufacturer) with no commensurate safety benefit".

#### **2.2.1 Safe Life Inspection Requirements**

Since the accident hub was certified using safe life criteria, no fixed interval inspections were required by regulation to ensure continued airworthiness. The Pratt and Whitney Overhaul Manual (OHM) specifies only "opportunistic inspections", consisting of FPI to detect "large flaw" defects during shop visits. The accident hub was inspected only once, by Delta using FPI, during a single shop visit after 12,693 flight cycles<sup>26</sup>. An existing fatigue crack, extending 0.9 inches into the tie bolt hole, and 0.46 inches across the aft face of the hub adjacent to the tie bolt hole, was not detected during the single inspection opportunity.

#### **2.2.2 Safe Life Certification Assumptions**

The 20% life reduction used to determine hub safe life limit (20,000 cycles) does not account for any initial material or manufacturing induced flaws in the hub, i.e. flaws that are not present in the fatigue test sample coupons. Merely increasing the hub design ultimate stress does not guarantee an increase in its safe life unless the life reducing effects of any initial flaws are filly accounted for. The hub design and continued airworthiness certification criteria, based on FAR 33.14 and 33.4 respectively, do not account for the effect of these flaws on fatigue life.

Metallurgist's Factual Report *<sup>25</sup>*

*<sup>26</sup>*NTSB **Docket** No. **SA-515,** Exhibit 110

<span id="page-20-0"></span>For example, the existing **-219** fan hub safe life cycle limit ("Ll") is predicated on takeoffs at normal take-off thrust. Based on FAR **33.14,** it should be necessary to determine a design minimum fatigue life **("L2")** predicated on start-stop cycles at *maximum power and thrust ratings.* 

The determination of design minimum fatigue life should also consider the scatter factor associated with the **S-N** (stress life prediction) curve of the hub material. The fatigue life scatter factor does not take into account the variations in the manufacturing quality of the hub, and the possible existence of induced 'initial' flaws. Currently it is assumed that initial flaws will be detected during the manufacturer's inspection process. There is no provision made for flaws that may pass undetected, or that may be detected but then not be recognized as significant. Currently there are no provisions for characterizing manufacturing flaws in terms of location, size, orientation and distribution. In calculating a reliable safe life, such characterizations should be quantified into an *initial fatigue quality parameter*<sup>27</sup> that accounts for growth characteristics of the initial flaw population, and consequently causes a firther reduction to minimum fatigue life **("L3").** Initial flaws can typically be grouped into three categories: (1) defects during initial forming, such as voids and inclusions, *(2)* random flaws induced during normal machining and handling, such as sharp notches, and **(3)** improper heat treatment or machining-induced flaws in the material's microstructure in the form of inhomogenities and cracks. In the accident hub, cracks induced from an improper machining operation and altered microstructure contributed to the failure.

**A** safe life criterion that includes initial flaws (e.g. cracks, notches) that have passed undetected into service implies that the crack growth rate in fatigue must initially be stable, predictable, and the life of the part must be exceeded before the crack becomes unstable. In the accident fan hub, a fatigue striation count on the fracture surface indicated that the crack growth was initially stable till around 9000 cycles. After that it became unstable and finally failed around 14,000 cycles, as shown in [Figure](#page-30-0) **3.** Theoretical predictability of crack growth appears possible based on an initial analysis using **NASGR028** material data for the Ti-6Al-4V alloy and crack growth equations, and estimated maximum cyclic stress. The size of the initial flaw in the accident hub was taken to be the depth of the high alpha content and high hardness microstructure found in the smeared area of the tie bolt hole surface, and measured as 0.017 inches. Based on fractography the crack was assumed to have been a corner crack with an aspect ratio of **2.35.** Using these initial flaw sizes, the predicted crack size as a finction of fatigue cycles was plotted using **NASGRO.** [Figure](#page-30-0) **3** shows that there is an excellent correlation between the theoretical and actual crack growth rate. This correlation therefore suggests that analytical determination of cycles-to-failure can be obtained with good accuracy if the initial damage/flaw is known and properly characterized.

For cases where the initial flaw size can be characterized as a short fatigue crack whose initial growth rates are not well defined,<sup>29</sup> additional work characterizing growth rates must be accomplished. The cycles to transition from a short crack to a crack that can be described by

**Appendix** B, **3 27** 

**Appendix B, 4 28** 

*<sup>29</sup>***Appendix B, 5, 6** 

linear elastic fracture mechanics should be determined. This is necessary to correctly assess the impact of the distribution of crack sizes likely to be found on the total fatigue life of the hub.

Thus, characterization of initial flaws in a hub after manufacturing will allow prediction of the minimum life to fatigue failure ("L3") based on crack growth considerations. In other words, the "L3" life is the damage tolerance based limit for Paris Law behavior, during which the crack growth rate is predictable. "L3" can be considered equivalent to the "initial service life" that had been proposed, but not accepted, by the commenter during the FAR 33.14 review. At that time no quantitative means of establishing this initial life was proposed. A clearly defined, quantitative approach, is proposed in this report that aims at removing any arbitrariness in defining the initial life "L3".

#### **2.2.3 Continued Airworthiness**

Continued airworthiness is contingent upon the enactment and application of minimum certification standards, and also the interdependence of the original equipment manufacturer delivering a "defect free" part that meets the type design certification basis, coupled with the operator performing inspections during the service of the life of the part that will detect rogue flaws, accidental damage and environmental damage.

Current continued airworthiness certification of the -2 19 fan hub does not include a schedule of inspection intervals based on the application of damage tolerance methodology.

There is also no provision for identification of the critical stressed area of the part coupled with a characterization of initial flaw size or material quality index. The material quality index must be linked to the manufacturer's ability to detect manufacturing defects of the type and extent experienced on the accident hub. The material quality index must not be dependent on the subjective interpretation of visual inspection techniques such as BEA, but instead must be characterized by the use of new high frequency eddy current inspection techniques capable of detecting the types of titanium microstructure deficiencies found during the investigation of this accident (see Section 1.5.1). The current certification basis of the -219 fan hub does not include a fixed interval inspection schedule based on analytically predicted stable growth rates (validated by tests) of low cycle fatigue cracks that emanate from rogue manufacturing flaws .

Current continued airworthiness certification does not require the application of supplemental non-destructive testing methods, such as high frequency eddy current inspection, for the critical stressed areas (tie bolt holes) of the fan hub. Under current certification guidelines, the same "large flaw" inspection technique, i.e. FPI, is used to detect damage in all areas of the hub, including both high and low stress areas.

These issues are considered inadequacies in the current safe life continued airworthiness certification process.

#### <span id="page-22-0"></span>**2.2.4 Damage Tolerance Criteria**

The application of damage tolerance criteria provide several advantages (as opposed to a reliance on safe life limits) to the effective management of continued airworthiness during the service life of the part. A damage tolerance continued airworthiness certification program:

- Requires that the reliability of inspection procedure is quantified (example 90% probability of detection with a 95% confidence level).
- Increases confidence in being able to predict both the rate of growth of a small defect  $\bullet$ (crack), and the critical crack size at which failure will occur under specified loads.
- Provides repeat inspection opportunities to detect a crack prior to failure (required to achieve 95% confidence level).
- Addresses initial material quality (such as microstructure damage present in the accident  $\bullet$ fan hub), defined as a distribution of intrinsic microstructural anomalies that might lead to failure.30 This concept quantifies the effect of defects below designated **NDE** capability.
- Requires more analysis and testing (enhanced reliability) than the low cycle fatigue based  $\bullet$ systems certified under safe life requirements.
- Less scatter (enhanced reliability) in the measurement of fracture mechanics properties  $\bullet$ than in low cycle fatigue life determination.
- $\bullet$ Failure prediction is based on the presence of damage (material defects, manufacturing defects, environmental and accidental damage).

All of these aspects combine to yield inspection size versus remaining life predictions for specific material and applications, which in total form a quantitative basis of damage tolerant design and life prediction approach.

#### **2.2.5 Certification Requirements** - **Summary**

Safe Life determination using life reduction factors must be rigorously substantiated by analysis and test. In addition, life reduction factors must account for manufacturing defects, material defects, accidental damage and environmental damage that may go undetected in the part's service life. Research is needed that establishes validation criteria for the use of life reduction factors that can be applied to safe life certification.

Not all factors affecting fatigue crack initiation (manufacturing defects, material defects, accidental and environmental damage) can be accounted for in the service life of critical engine rotating parts. Initial fatigue design data is subject to considerable statistical variability. Significant scatter is observed in published S-N fatigue data. Conventional safe-life design acknowledges statistical scatter qualitatively through the use of life reduction factors, but it does not quantify the resulting level of reliability<sup>31</sup>. Safe life certification must be supplemented by continuous airworthiness limitations based on rigorously applied damage tolerance criteria<sup>32</sup>.

**Appendix** B, **8**  *30* 

**Appendix B, 9 31** 

**Appendix** B, **10, 11 32** 

<span id="page-23-0"></span>The damage tolerance analysis should account for manufacturing damage and material defects using an initial flaw assumption. Analysis should not be based on fatigue crack initiation (threshold credit). Additional research is needed to determine an equivalent initial crack size and shape, but validation of initial flaw size should be part of the continued airworthiness certification process.

#### **2.3 Operator Inspection Process**

Maintenance records indicate that the -219 fan hub had been cleaned, processed and subjected to fluorescent penetrant inspection in accordance with Delta's standard practices pertaining to titanium engine parts manufactured by Pratt & Whitney. Nevertheless, metallurgical analyses of the failed hub revealed that at the time of the last FPI in December 1995, the crack was present on the aft face of the hub  $(0.46$  inches) and along the wall of the tie bolt hole  $(0.9)$ inches); yet, the crack was not detected and the fan hub was returned to service. This calls into question two safety concerns with regard to FPI: the integrity of the existing cleaning and/or FPI processes, and the adequacy FPI as an inspection method for critical rotating parts.

The first concern focuses on the possibility that the cleaning or inspection process was compromised, resulting in a contamination of the region of the part that contained the crack. This explanation is bolstered by the fact that the metallurgical examination of the crack surface failed to detect the presence of trace elements of the fluorescent dye penetrant. Although it cannot be conclusively determined from this that no dye penetrated the crack during the FPI, the absence of trace elements on the surface of the crack in combination with the failure of the inspector to detect a crack nearly half an inch in length on the aft face of the hub suggests that it is highly unlikely that fluorescent dye had penetrated the crack during the FPI processing.

#### **2.3.1 Part Drying**

One possible source of contamination is the presence of water in the crack that would preclude penetration by the fluorescent dye. Delta's process standard for titanium parts utilizes a flash drying technique at the end of the alkaline cleaning procedure, except for certain Roll Royce parts<sup>33</sup>. This technique involves immersing the part in a vat of hot water until the temperature of the part equals the temperature of the water. When removed from the hot water, the part dries through evaporation. However, it is apparent from observation of the drying process and from the testimony of witnesses at the public hearing on this accident that there are no tools available to cleaning personnel for determining whether the temperature of the part equals the water temperature; or that the part has been adequately heated to ensure the evaporation of water that may be present in cracks or other surface defects. It is therefore possible that the part progressed beyond the flash drying stage of the cleaning process with water still present in the crack. Because dye penetrant is essentially oil-based, it will be repelled by any water present in the crack.

The cleaning personnel responsible for drying the part must rely on the prescribed water temperature and dwell time requirements for flash drying stated in the process standard,

**<sup>33</sup>**Rolls Royce Overhaul Manual

<span id="page-24-0"></span>followed by a visual inspection of the evaporation of water on the surface. It was noted during the public hearing that the recommended temperature and dwell times for flash drying were developed to facilitate the evaporation of water on the surface of the part and do not account for possible water present in deep cracks or other surface anomalies. There is no way for cleaning personnel to know whether or where a crack may exist in order to assess whether water has fully evaporated from the crack.

The use of oven drying has been raised as a possible solution to the deficiencies of flash drying. Of the three OEMs of titanium engine parts used by Delta, only Rolls Royce specifically requires oven drying of Group **A** parts and compressor and turbine blades. Nevertheless, whether a part is dried through flash drying or oven drying, *time* is the critical variable for ensuring the complete evaporation of water on the surface or in a crack. **A** single standard for determining the sufficient time at an elevated temperature is therefore needed whether the part is dried in an oven or flash dried.

#### **2.3.2 Plastic Media**

**A** second possible source of contamination is plastic media. Delta's process standard provides guidelines to cleaning personnel on the proper technique for blasting a part with plastic media to prevent the media from becoming entrapped or from masking potential defects on the surface of a part through peening or material displacement. **A** number of expert witnesses in the public hearing advocated rinsing the part after subjecting it to media blast to loosen and remove potential media contamination. Delta's process standard calls for blowing the part with compressed air following the media blast, and the recommendations from the OEM's differ on this point. Only Pratt & Whitney's recommended procedure refers to a pressure spray rinse to remove plastic media, "*if needed*"; however, no guidance is provided by Pratt & Whitney as to whether a rinse is needed or not. Neither Rolls Royce nor General Electric recommend the use of a spray rinse on titanium parts after plastic media blasting.

#### **2.3.3 OEM Guidance**

**As** noted by a number of expert witnesses in the Safety Board's public hearing on this accident, FPI is a highly process-dependent activity. That is, for FPI to be an effective tool for detecting and rejecting critical parts with flaws that exceed tolerance limits, cleaning, preparation and inspection procedures must be closely followed. The fact that there is inconsistency or disagreement in the various OEM recommended procedures for similar parts is therefore a serious concern. **A** single set of procedural standards for cleaning and inspecting similar parts should be developed. It should incorporate the best scientific information available for enhancing the effectiveness of the cleaning and inspection processes. It is recommended therefore, that a task force comprising representatives from government and the industry be convened to review the adequacy of existing original equipment manufacturer procedures for cleaning and processing critical parts prior to visual inspection, and develop standardized procedures for cleaning and processing.

#### <span id="page-25-0"></span>**2.3.4 FPI Effectiveness**

The second safety concern focuses on whether FPI is an effective inspection method for critical engine parts like the -219 hub; parts that are characterized by complex geometries or areas that are difficult to inspect visually, such as tie bolt holes. In this particular case, at the time of the 1995 inspection the crack had propagated on the aft face of the hub to a length that should have been detected through FPI. However, the crack initiated and was detectable in a tie bolt hole many cycles before it emerged on the aft surface. Had the FPI been conducted earlier (prior to its emergence on the aft face), it is likely that the probability of detecting the crack would have been much lower given the difficulties faced by inspectors in adequately conducting a *360"* visual inspection of deep holes. Although tools such as mirrors, boroscopes and pen lights are available to inspectors in the FPI tent, difficulties in inspecting the tie bolt and stress redistribution holes were noted by inspectors during the investigation, and were observed firsthand by investigators.

Following the accident, Delta worked with Pratt & Whitney and the FAA to develop an eddy current inspection technique for use as a supplemental inspection of the tie bolt and stress redistribution holes on the -219 hub. The eddy current procedure was adopted in an AD issued in February 1997 that mandates a combination of eddy current and FPI inspections at regular intervals throughout the safe life of the fan hub. There is concern that there are other critical rotating parts characterized by complex geometries or deep holes for which visual inspection techniques are not well suited. It is therefore recommended that a task force comprising government and industry representatives be convened to identify critical areas of component parts for which existing visual inspection procedures may be unsuited to detecting existing flaws, and develop alternative or supplemental inspection techniques that increase the probability of detecting such flaws to acceptable limits.

#### **3. CONCLUSIONS**

#### **3.1 Findings**

- 1. **A** severe microstructural deformation of the titanium on the wall of the tie bolt hole of the accident hub, adjacent to the origin of the fracture, was caused by abusive machining during the manufacturing process.
- 2. The non-destructive inspection and evaluation processes used during manufacturing were not effective in assessing the microstructural deformation on the part caused by abusive machining, and in making a determination to reject the part for service.
- *3.*  **A** change in the machining process that involved drill replacement (to a coolant channel drill) in the manufacture of JTSD-200 fan hubs was classified as an "insignificant change", and given engineering approval by the manufacturer without adequate research as to potential abusive effects of the replacement drill on the microstructure of the titanium.
- **4.**  High temperatures resulting from manufacturing machining processes, particularly drilling, have the potential for affecting, or changing, the microstructural characteristics of titanium alloys.
- *5.*  Relatively few studies have been conducted, and little data is available, that quantify the potential effects of machining, and drilling in particular, on the surface microstructure of titanium alloys.
- 6. Assumptions used to determine and certify the safe life limit for the JT8D-219 fan hub do not account for life reducing initial flaws that might exist when the part comes out of the manufacturing process.
- *7.*  The certification of the -219 fan hub does not include development of an initial fatigue quality parameter that accounts for growth characteristics of cracks that might develop as the result of manufacturing flaws.
- **8.**  Damage tolerance criteria are not included in the certification of the -219 fan hub; there is no requirement for any inspections that ensure continued airworthiness of the part to the life limit.
- *9.*  The procedures that cover the cleaning, processing, and inspection of a given part manufactured by one original equipment manufacturer (OEM) may differ from the procedures that cover a similar part manufactured by other OEMs.
- 10. Cleaning personnel rely on a visual inspection of the evaporation of water on the surface of the part to determine that a part is adequately dry. There is no published guidance or tools available to the cleaner to determine whether water that may be present in cracks or other surface anomalies has fully evaporated prior to the application of fluorescent penetrant dye.
- 1 1. During two observations of FPI inspections, it was observed that an ultraviolet spotlight does not effectively penetrate into the tie bolt holes when viewed from the top of the hub. From the bottom of the hub, proper alignment can only be done if the inspector rotates the hub under a particular frame of reference. One inspector said that finding cracks in the bolt and balance weight holes is not really possible. **A** second inspector also stated to investigators that it was very difficult to see into the holes on the hub.

**BATAIN**TTT

#### **3.2 Probable Cause**

The probable cause of this accident was the catastrophic failure of the left engine first stage fan hub as a result of a low cycle fatigue crack that originated from a microstructure defect created by abusive machining during the manufacturing process. Contributing to this accident was:

- the failure of the manufacturer to adequately evaluate changes to the approved drilling  $\bullet$ process, and the effects of those changes on the microstructure of titanium,
- the failure of the manufacturer to adequately assess visual abusive machining damage indications, and to effectively utilize a non-destructive method to evaluate these findings,
- $\bullet$ the absence of an effective standard for rejecting parts with known abusive machining defects detected through Blue Etch Anodize inspection methods during the manufacturing process,
- the failure of the certification process to adequately account for factors that can preclude  $\bullet$ critical parts from reaching their safe life limit,
- the absence of standardized guidance for air carriers for cleaning and processing of critical  $\bullet$ parts during the fluorescent penetrant inspection process used by operators, and
- the absence of an effective visual inspection method suitable for detecting flaws in critical  $\bullet$ rotating parts with complex geometries and deep holes.

#### $\overline{\mathbf{4}}$ . **RECOMMENDATIONS**

**As** a result of the findings from this investigation, it is recommended that:

- 1. To the FAA: Convene an industry task force to identify critical machining processes, especially drilling, and associated potential deleterious effects on the microstructure of titanium; and recommend requirements for the evaluation of these practices, and subsequent changes to certified tools and methods.
- 2. To the FAA: Sponsor an industry/academia research effort to develop additional nondestructive testing methods, such as the new application of high frequency eddy current inspection, that will detect defective microstructure surface layers (interstitial oxygen stabilized alpha) in critical rotating engine parts; and to establish a certification requirement for the reliable detection of such damage.
- 3. To the FAA: Charter an industry task group to review and recommend revisions to current certification procedures for safe life engine rotating parts to require the validation of life reduction factors, or as an alternative, to establish a minimum fatigue probabilistic reliability standard to ensure that premature failure is extremely remote; and to require that

**MARINE CA** 

continued airworthiness of safe life parts be managed by damage tolerance criteria as a supplement to safe life determination.

- 4. To the FAA: Convene an industry/government task force to review the adequacy of existing original equipment manufacturer procedures for cleaning and processing critical parts prior to visual inspection, and develop standardized procedures for cleaning and processing.
- 5. To the FAA: Convene an industry/government task force to identify critical areas of component parts for which existing visual inspection procedures may be unsuited to detecting existing flaws, and develop alternative or supplemental inspection techniques that increase the probability of detecting such flaws to acceptable limits.

**THE MODEL CAR** 

<span id="page-29-0"></span>

Figure 1. Micrograph of the near-surface region of an oxygen contaminated Ti-6Al-4V alloy, showing a large amount of alpha phase (white areas). [Appendix A. Ref. 1].



**[Figure](#page-3-0)** 2. Thickness of the oxide layer formed on Ti-6Al-4V alloy as a function of time at different temperatures [Appendix A, Ref. 1].

<span id="page-30-0"></span>



Page 26

#### **APPENDIX A**

#### **References** - **Microstructure Damage from Machining**

- 1. Wood and R. J. Favor, Titanium Alloys Handbook. MCIC-HB-02, Section I, sponsored by Air Force Materials Laboratory, Wright-Patterson Air Force Base, OH, Dec. 1972.
- *2.* Norman Zlatin and Michael Field, "Machinability parameters on new and selective aerospace materials", USAF Technical Report AFML-TR-7 1-95, Metcut Research Associates Inc., Cincinnati, OH, 1971.
- 3. Norman Zlatin, Michael Field and William P. Koster, "Machining of new materials", USAF Technical Report AFML-TR-67-339, Metcut Research Associates Inc., Cincinnati, OH, 1967.
- 4. "Microstructure and morphology of surface oxide films on Ti-6Al-4V", Journal of Materials Research 5, 8, (1990), p 1662.
- 5. Pratt & Whitney Research Report 704297-2440, Docket No. SA-515, Exhibit No. 8D, National Transportation Safety Board, Washington, DC, April 5, 1982.
- 6. Metals Handbook: Vol. 1. Properties and Selection of Metals, Taylor Lyman, ed., 8th Edition, American Society for Metals, Metals Park, OH, 1961, p 538.
- 7. Metals Handbook: Vol. 2. Heat Treating. Cleaning and Finishing, Taylor Lyman, ed., 8th Edition, American Society for Metals, Metals Park, OH, 1961, p 301.
- **8.** Papers presented in Titanium 80, Proceedings of the 4th International Conference on Titanium, Vols. 1-4, H. Kimura and 0. Izumi, eds., TMS-AIME, Warrendale, **PA,** 1980.
- 9. Margolin, J. C. Williams, J. C. Chesnutt and G. Leutjering, " A review of the fracture and fatigue behavior of Ti alloys", *ibid,* Vol. I, p 169.
- 10. Peters, A. Gysler and G. Leutjering, "Influence of microstructure on the fatigue behavior of Ti-6Al-4v", *ibid,* Vol. 3, p 1777.
- 1 1. Broichhausen and M. Telfah, "Influence of manufacturing method and surface condition on the fatigue strength of Ti-6AI-4V material", *ibid,* Vol. 3, p 1797.
- 12. Krishnamohanrao, V. V. Kutumbarao, and P. Ramarao, "Fracture mechanism maps for titanium and its alloys", Acta Metallurgica 34, 9, (1986), **p** 1783.
- 13. Imam and D. M. Gilmore, "Fatigue and microstructural properties of quenched Ti-6Al-4V", Metallurgical Transactions 144 (1983), **p** 233.

**Material Control** 

**Appendix A** Page 1 **of** 2

- 14. " Effects of the alpha/beta phase proportion on the superplasticity of T-6Al-4V and iron modified Ti-6Al-4V", Materials Science and Engineering A: Structural Materials, Properties Microstructure and Processing A154, 2, (1992), p 165.
- 15. Wood and R. J. Favor, Titanium Alloys Handbook, MCIC-HB-02, Section 3, sponsored by Air Force Materials Laboratory, Wright-Patterson Air Force Base, OH, Dec. 1972.
- 16. Kahles, M. Field, D. Eylon and F. H. Froes, "Machining of Titanium", Journal of Metals, **(1985),** p 27.
- 17. Bhaumik et al., "Machining Ti-6AI-4V alloy with a wBN-cBN composite tool", Materials and Design 16, **4,** (1995), p 221.
- 18. Low Stress Grinding, Publication No. MDC 83-103, Metcut Research Associates Inc., Machinability Data Center, Cincinnati, OH, 1970.
- 19. Norman Zlatin and Michael Field, "Machinability parameters on new and selective aerospace materials", USAF Technical Report AFML-TR-69- 144, Metcut Research Associates Inc., Cincinnati, OH, 1969.
- 20. Wood and R. J. Favor, Titanium Alloys Handbook, MCIC-HB-02, Section I, sponsored Jeelani and K. Ramakrishnan, " Surface damage in machining titanium -6AI-2Si-4Zr-2Mo alloy", Journal of Materials Science 20, (1989, p 3245.
- *2* **1.** Volvo Manufacturing Drawing 0- 169282, Volvo Hub Front Drill Process History, Docket No. SA-5 15, Exhibit No. 8B-1, National Transportation Safety Board, Washington, DC.
- 22. Powerplants Group Chairman's Factual Report, Rev. 1, Docket No. **SA-5** 15, Exhibit No. SA, National Transportation Safety Board, Washington, DC, March 7, 1997.
- 23. ANE-180 "Evaluation Report of Pratt and Whitney Quality Systems" (PWA-370, "Engineering Source Approval"), Docket No. SA-5 15, Exhibit No. SH, Attachment **2,**  Phase **I1** Report (Volvo), National Transportation Safety Board, Washington, DC, Sept. 30, 1996.
- 24. Maintenance Group Chairman's Factual Report: English Translation of Volvo's Manufacturing Records on Hub **S/N** 32971, Docket No. SA-5 15, Exhibit No. 11 E, National Transportation Safety Board, Washington, DC.

**landin** 11

25. ANE-180, "Evaluation Report of Pratt and Whitney Quality Systems", Phase I Report, Docket No. SA-5 15, Exhibit No. **8** G, National Transportation Safety Board, Washington, DC, Sept. 30, 1996.

**Appendix A** 

 $\sim$   $\sim$ 

**Page** *2* of 2

#### **APPENDIX B**

#### **References** - **Part Certification**

- **1.** Stephen Pearlman, Sr. Metallurgist, Pratt & Whitney, private communication during NTSB Metallurgy Team Investigation of JT8D-219 PNS Accident at NTSB Hdq., Washington, D.C., July 8 - 12, 1996.
- 2. Metallurgist Factual Report 96-13 1, Exhibit 15-A, NTSB Docket No. SA 515.
- 3. L. Rudd, **J.** N. Yang, S. D. Manning and B. G. W. Yee, "Probabilistic fracture mechanics analysis methods for structural durability", AGARD CP-328, Behaviour of Short Cracks in Airframe Structural Components, 55th Meeting of the AGARD Structures and Materials Panel, Toronto, Canada, Sept. 1982.
- 4. G. Forman et al., Fatigue Crack Growth Computer Program "NASAlFLAGRO" Version 2.0, NASA, JSC-22267A, Houston, TX, 1994.
- 5. Drexler and J. Statecny, "Short crack problems in gas turbine disks", from Advances in Fracture Research, Vol. *5,* S. R. Valluri et al., eds., 6th Int. Conf. on Fracture, N. Delhi, India, Dec. 1984.
- *6.* Nemec and **J.** Drexler, "Short crack development in mechanical structures", from Advances in Fracture Research, Vol. 5, S. R. Valluri et al., eds., 6th Int. Conf. on Fracture, N. Delhi, India, Dec. 1984.
- 7. D. Rummel and G. Matzkanin, The Nondestructive Evaluation (NDE) Capabilities Handbook, Nondestructive Testing Information Analysis Center publication number DB-95-02, Austin, TX.
- **8.** E. Farmer and M. **C.** Vanwanderham (Pratt & Whitney), "Damage tolerance concepts for advanced materials and engines", AGARD CP-449, Application of Advanced Materials for Turbomachinery and Rocket Propulsion, papers presented at the 72nd spealists meeting of the Propulsion and Energetics Panel, Bath, U.K., Oct. 1988, p 6-1.
- 9. Barry S. Spigel, "Safe life reliability design for rotorcraft", J. American Helicopter Society - 36, 1, (1991) **p** 78.
- 10. 0. N. James, "The use of reliability techniques in civil aircraft structural airworthiness <sup>a</sup> CAA view", The Aeronautical Journal *92,* 91 1, (1988), **p3.**
- 11. E. W. Stone, "The reliability of inspection techniques in relation to damage tolerant design", The Aeronautical Journal 92, 911, (1988), p5.

سياسيا والمراج

## **APPENDIX** *C*

 $\label{eq:1.1} \frac{1}{2}\int_{\mathbb{R}^{3}}\left|\frac{1}{\sqrt{2}}\left(\frac{1}{\sqrt{2}}\right)^{2}+\frac{1}{2}\int_{\mathbb{R}^{3}}\left|\frac{1}{\sqrt{2}}\left(\frac{1}{\sqrt{2}}\right)^{2}+\frac{1}{2}\int_{\mathbb{R}^{3}}\left|\frac{1}{\sqrt{2}}\left(\frac{1}{\sqrt{2}}\right)^{2}+\frac{1}{2}\int_{\mathbb{R}^{3}}\left|\frac{1}{\sqrt{2}}\left(\frac{1}{\sqrt{2}}\right)^{2}+\frac{1}{2}\int_{\mathbb{R}^{3}}\$ 

 $\frac{1}{2}$  and  $\frac{1}{2}$  a

 $\mathcal{L}^{\text{max}}_{\text{max}}$ 

## **PROCESS COMPARISON DELTA vs. OEM's CLEANING AND BLASTING**

#### **PROCESS OVERVIEW**

 $\ddot{\phantom{0}}$ 

The purpose of this chart is to illustrate the procedures performed by Delta on a titanium part compared to what is specified by the original equipment manufacturers (OEM's). The **blue** text indicates differences.



*0* This disk is sent for outside repair (not associated with the differences identified).

## **PROCESS COMPARISON DELTA vs. OEM's CLEANING AND BLASTING**

## **PARTS PROCESSING FLOW CHART**

**The purpose of this chart is to illustrate the processing sequence of titanium parts requiring dry film lubricant removal** per **P.S. 900-1-1 No. 18.** 



**APPENDIX C** 5/22/97 **Page 2 of 11** 

the company

## PROCESS **COMPARISON DELTA vs. OEM's**  PRECLEANING - AQUEOUS

The charts that follow illustrate the differences between Pratt & Whitney's, General Electric's and Rolls Royce's procedures and Delta's replacement procedures.





APPENDIX C 5/22/97 [Page](#page-7-0) **3** of 11

 $\frac{1}{4}$ 

## **PROCESS COMPARISON DELTA vs. OEM's PRECLEANING** - **AQUEOUS**



**Page 4 of 11 APPENDIX C** *5122/97* 

 $\sim$ 

 $\mathcal{A}$ 

 $\frac{1}{2}$ 

**BETWEEN** 

 $\bar{z}$  $\frac{1}{2}$ 

## **PROCESS COMPARISON DELTA vs. OEM's PRECLEANING** - **AQUEOUS**



 $\frac{1}{2}$ 

 $\mathbb{C}$ 

 $\blacksquare$ 

## **PROCESS COMPARISON DELTA vs. OEM's**  CLEANING - ALKALINE





 $\frac{1}{2}$ 

Ť,

## **PROCESS COMPARISON DELTA vs. OEM's CLEANING** - **ALKALINE**



Ĵ.

 $\frac{1}{2}$ 

## **PROCESS COMPARISON DELTA vs. OEM's PLASTIC MEDIA BLASTING**



 $\frac{1}{2}$  .

İ.

## **PROCESS COMPARISON DELTA vs. OEM's PLASTIC MEDIA BLASTING**



 $\begin{array}{c} 1 \\ 1 \\ 2 \end{array}$ 

 $\mathbb{C}$ 

## **PROCESS COMPARISON DELTA vs. OEM's PLASTIC MEDIA BLASTING**

<span id="page-44-0"></span>

References: Pratt & Whitney's Standard Practice, 70-2[1 -00, SPOP 18,](#page-22-0) [19](#page-23-0) & 209 Rolls Royce's 594J Engine Overhaul Manual, 70-00-00, O.P. 102, 104 & 136 General Electric's Standard Practices, 70-21-00, Method 04, 09, & 22 Delta Air Line's Process Standard Manual, 900-1 [-1 No. 18](#page-22-0) [& 21](#page-25-0)

## **PROCESS COMPARISON DELTA vs. OEM's SHELL** BLASTING

This process comparison is included due to shell blasting being accomplished in lieu of plastic media blasting on Roll Royce's fan disc.



 $\epsilon$ 

 $\frac{1}{2}$ 

 $\begin{array}{c} \begin{array}{c} \begin{array}{c} \begin{array}{c} \end{array} \\ \begin{array}{c} \end{array} \end{array} \end{array} \end{array}$ 

## **APPENDIX D**

 $\mathcal{L}^{\mathcal{L}}$  and  $\mathcal{L}^{\mathcal{L}}$  are the contribution of the set of the contribution of  $\mathcal{L}^{\mathcal{L}}$ 

 $\sim$  case

 $\ddot{\phantom{a}}$ 

 $\label{eq:2.1} \mathcal{L}(\mathcal{L}^{\text{max}}_{\mathcal{L}}(\mathcal{L}^{\text{max}}_{\mathcal{L}}(\mathcal{L}^{\text{max}}_{\mathcal{L}}(\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}}(\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}_{\mathcal{L}^{\text{max}}$ 



 $\mathbf{r}$ 

 $\begin{array}{c} \hline \end{array}$ 



 $\mathbb{I}$ 

 $\overline{\phantom{a}}$ 





 $\begin{array}{c} \begin{array}{c} 1 \\ 1 \end{array} \\ \begin{array}{c} 1 \end{array} \end{array}$ 

 $\label{eq:1.1} \mathcal{L} = \mathcal{L} \left( \mathcal{L} \right) \mathcal{L} = \mathcal{L} \left( \mathcal{L} \right) \mathcal{L} \left( \mathcal{L} \right)$ 

